



Structural and Optical Properties of Silicon Nanowire Arrays Fabricated by Metal Assisted Chemical Etching With Ammonium Fluoride

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Here we report on the metal assisted chemical etching method of silicon nanowires (SiNWs) manufacturing, where the commonly used hydrofluoric acid (HF) has been successfully replaced with ammonium fluoride (NH₄F). The mechanism of the etching process and the effect of the pH values of H₂O₂: NH₄F solutions on the structural and optical properties of nanowires were studied in detail. By an impedance and Mott-Schottky measurements it was shown that silver-assisted chemical etching of silicon can be attributed to a facilitated charge carriers transport through Si/SiO_x/Ag interface. It was shown that the shape of nanowires changes from pyramidal to vertical with pH decreasing. Also it was established that the length of SiNW arrays non-linearly depends on the pH for the etching time of 10 min. A strong decrease of the total reflectance to 5–10% was shown for all the studied samples at the wavelength <800 nm, in comparison with crystalline silicon substrate (c-Si). At the same time, the intensities of the interband photoluminescence and the Raman scattering of SiNWs are increased strongly in compare to c-Si value, and also they were depended on both the length and the shape of SiNW: the biggest values were for the long pyramidal nanowires. That can be explained by a strong light scattering and partial light localization in SiNWs. Hereby, arrays of SiNWs, obtained by using weakly toxic ammonium fluoride, have great potential for usage in photovoltaics, photonics, and sensorics.

Keywords: silicon nanowires, impedance, total reflectance, photoluminescence, Raman scattering

INTRODUCTION

In recent decades, the possibility of using silicon nanowires (SiNWs) in sensorics (Cui et al., 2001; Wang and Ozkan, 2008; Cao et al., 2015; Georgobiani et al., 2018), photovoltaics (Kelzenberg et al., 2008; Stelzner et al., 2008; Sivakov et al., 2009), photonics (Brönstrup et al., 2010), and micro-and optoelectronics (Föll et al., 2010; Yang et al., 2010) has been shown. Nanowires are usually obtained as a result of anisotropic growth of a 1D crystal on a nanometer scale. The first SiNWs were fabricated via bottom-up approach by vapor-liquid-solid (VLS) method with different noble metals (mostly gold) as catalyst (Wagner and Ellis, 1964). Metal-assisted chemical etching (MACE) of silicon was observed for the first time in the 1990s, when a cleaning solution HF-H₂O₂-H₂O was used to remove metal impurities from the silicon substrate (c-Si) (Morinaga et al., 1995). Then this method was adapted for luminescent porous silicon formation (Gorostiza et al., 1999;

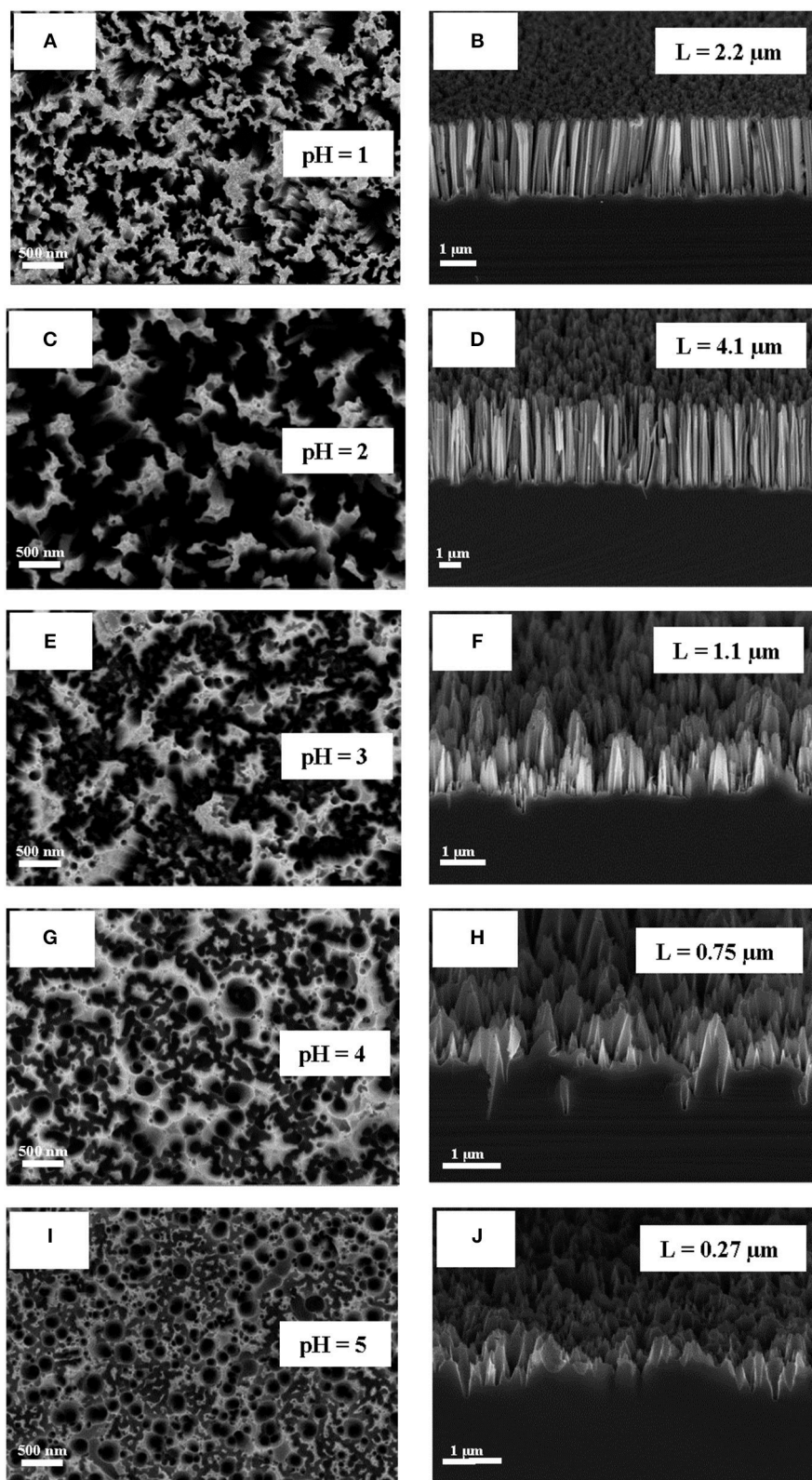


FIGURE 1 | (A,C,E,G,I) SEM micrographs of SiNWs with different pH of $\text{H}_2\text{O}_2:\text{NH}_4\text{F}$ (view from above); **(B,D,F,H,J)** SEM cross-sectional micrographs of SiNWs with different pH of $\text{H}_2\text{O}_2:\text{NH}_4\text{F}$.

Li and Bohn, 2000; Chattopadhyay et al., 2002). In 2002, Peng et al. for the first time adapted it for high aspect ratio SiNWs fabrication and systematically investigated the mechanism and further develop it into a new micro/nanofabrication method (Peng et al., 2002, 2006, 2008). Also MACE method of SiNWs fabrication was systematically investigated in Nahidi and Kolasinski (2006), Sivakov et al. (2010), Bai et al. (2012), and Dawood et al. (2012). Usually in MACE such catalysts, as nanoparticles of Au (Li and Bohn, 2000; Dawood et al., 2012), Ag (Sivakov et al., 2010), or Pt (Li and Bohn, 2000; Chattopadhyay et al., 2002) and such oxidizing agents as H_2O_2 (Li and Bohn, 2000; Sivakov et al., 2010; Dawood et al., 2012), KMnO_4 (Bai et al., 2012; Jiang et al., 2017), or $\text{Fe}(\text{NO}_3)_3$ (Nahidi and Kolasinski, 2006), are used in the process. SiNWs, which were fabricated by a standard MACE procedure, are found to possess such optical properties as extremely low total reflection (Gonchar et al., 2012), enhancement of Raman scattering and interband photoluminescence (PL) (Gonchar et al., 2014). However, HF, that is surely used in the MACE, is toxic and dangerous, and may also lead to hypocalcemia and hypomagnesemia (Bertolini, 1992). Therefore, it is very important, with a view to the future large-scale production of SiNWs, to study the possibilities of modifying the MACE method using safer and less toxic chemicals.

It is well-known that aqueous solutions of ammonium fluoride (NH_4F) can be used to dissolve SiO_2 , and the etching rate depends on the concentration of NH_4F and the pH of the solutions (Judge, 1971). Thus, NH_4F is shown can be used as an alternative to HF in the method of electrochemical etching in the manufacture of porous silicon, and the structural properties of the resulting porous silicon depend on the pH of the NH_4F solution used: at $\text{pH} = 4.5$ a pebble-like surface was formed, and at lower PH a nanoporous silicon layers were formed (Dittrich et al., 1995). Recently, the possibility of using NH_4F in the MACE process has been also shown, and optical properties of SiNW, formed using NH_4F , differed little from nanowires formed by

standard MACE technology with HF (Gonchar et al., 2016). However, the mechanism of the etching process and the influence of the pH of the etching solution on the structural and optical properties of SiNW remain open.

In this work, the etching process mechanism and the effect of pH values of $\text{H}_2\text{O}_2:\text{NH}_4\text{F}$ solutions on the structural and optical properties of SiNWs were studied using impedance measurements and Mott-Schottky analysis, as well as total reflectance, interband photoluminescence and Raman scattering intensities measurements.

METHODS

The samples of SiNWs were produced by MACE of (100)-oriented p-type c-Si wafer with resistivity of $10\text{--}20\ \Omega\cdot\text{cm}$. HF was replacement on NH_4F in all reactions. The PH value was

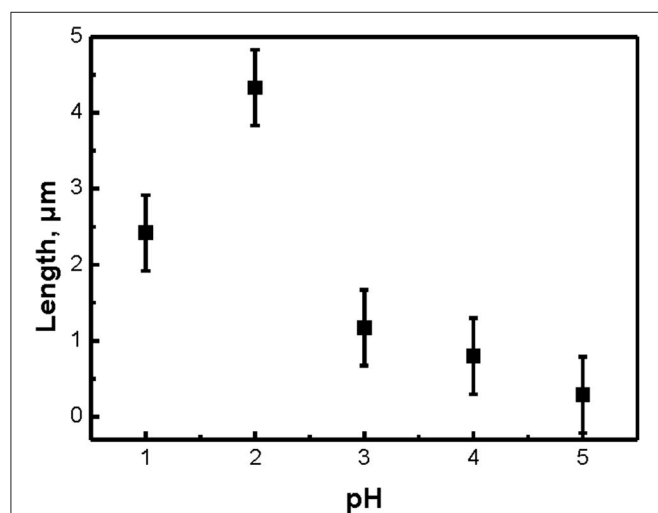


FIGURE 2 | The dependence of the length of SiNWs with different pH value of $\text{H}_2\text{O}_2:\text{NH}_4\text{F}$.

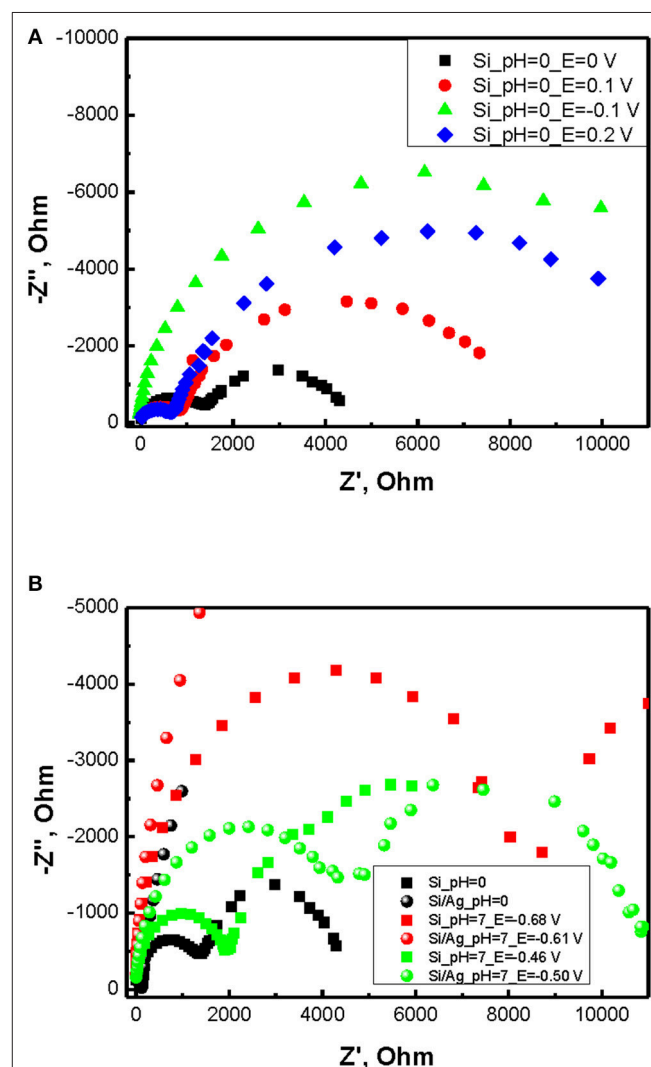
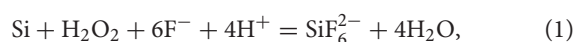


FIGURE 3 | (A) Impedance spectra of p-doped silicon in $\sim 5\text{M}$ $\text{NH}_4\text{F}/1\text{M}$ H_2SO_4 electrolyte containing 30% of H_2O_2 . **(B)** impedance spectra for different interfaces and pH.

controlled by PH indicator. Prior to the MACE procedure, c-Si wafers were rinsed in 2% HF solution for 1 min to remove a native silicon oxide. In the first stage of MACE process, c-Si wafers were placed in the aqueous solution of 0.02 M of silver nitrate (AgNO_3) and 5 M of NH_4F in the volume ratio of 1:1 for 30 s and a thin (~ 100 nm) layers of Ag nanoparticles were deposited on the surface of the wafers. In the second stage, c-Si wafers with Ag nanoparticles were placed in the etching solution containing 5 M of NH_4F and 30% H_2O_2 in the volume ratio of 10:1 for 10 min. The PH value of the NH_4F aqueous solution was changed by adding of H_2SO_4 droplets and varied in the range from 1 to 5. All the etching stages were carried out at room temperature. After the etching process all the samples were rinsed in de-ionized water and dried at room temperature. The main etching reaction the same that was described in Zhang et al. (2008):



however in our case the ions of F^- and H^+ were obtained not from the dissociation of HF as in standard MACE procedure, but from the dissociation of NH_4F and H_2SO_4 . Ag nanoparticles played the role of catalysts for the etching process. The removal of Ag nanoparticles from SiNW arrays was performed by immersing in concentrated (65%) nitric acid (HNO_3) for 15 min.

The structures of SiNWs were studied by a scanning electron microscope (SEM) of Carl Zeiss SUPRA 40 FE-SEM. Impedance spectra and Mott-Schottky measurements were performed using Solartron 1287 electrochemical interface and Solartron 1255B frequency response analyzer. All the measurements were carried out in three-electrode teflon cell using Ag/AgCl reference electrode joined through polypropylene Luggin capillary. The total reflectance (which includes both diffuse and specular components) spectra at the wavelength from 250 to 850 nm were studied with an integrating sphere on a Perkin Elmer spectrometer Lambda 950. The interband PL and Raman spectra were measured in a back scattering geometry with a Fourier-transform infrared (FTIR) spectrometer of Bruker IFS 66v/S equipped with a FRA-106 unit. Excitation was carried out by cw Nd:YAG laser at the wavelength $1.064 \mu\text{m}$ (excitation intensity was 100 mW and spot size was 2 mm). All experiments were carried out in air at room temperature.

TABLE 1 | Open circuit potential (OCP) and flat band potential for different interfaces and pH.

Sample	Open circuit potential, V	Flat band potential, V
Si_pH = 0–1	−0.17	−0.26
Si_pH = 2–3	−0.25	−0.34
Si_pH = 4–5	−0.34	−0.46
Si_pH = 6–7	−0.55	−0.75
Si/Ag_pH = 0–1	−0.14	−0.32
Si/Ag_pH = 2–3	−0.30	−0.38
Si/Ag_pH = 4–5	−0.33	−0.39
Si/Ag_pH = 6–7	−0.42	−0.50

RESULTS AND DISCUSSION

Typical SEM microphotographs of SiNW layers, which were obtained by using different pH of the etching solution $\text{H}_2\text{O}_2:\text{NH}_4\text{F}$ are presented in **Figure 1**. Note, that for pH = 6 or 7 the etching rate was very slow and the optical properties of SiNWs are slightly different from c-Si substrate. It is seen from the **Figure 1**, that the shape of SiNW is changing from vertical cylinders to pyramidal like structures with pH increasing. **Figure 2** presented the dependence of the length of SiNWs from the pH value. The length of SiNW is maximum at pH = 2 and then decreases with increasing pH. SiNW porosity was calculated by using Bruggeman model (Bruggeman, 1935) and was approximately 50–60% for all samples.

Impedance spectra of p-doped silicon in 5M $\text{NH}_4\text{F}/1\text{M}$ H_2SO_4 electrolyte containing 30% of H_2O_2 illustrate two semicircles with series resistance close to zero (**Figure 3A**). Thus, an equivalent circuit for the cell can be represented by parallel RC circuits connected in series. Applying positive bias vs. open circuit potential (OCP) leads to a first element resistivity decrease while increasing the radius of the second semicircle. Applying negative potential leads to first semicircle radius growth. As soon as Warburg resistance can be considered negligible in concentrated $\text{NH}_4\text{F}/\text{H}_2\text{SO}_4$ solution, the presence of the second semicircle can be referred to an electric double layer with non-equilibrium silicon oxide formed at the surface

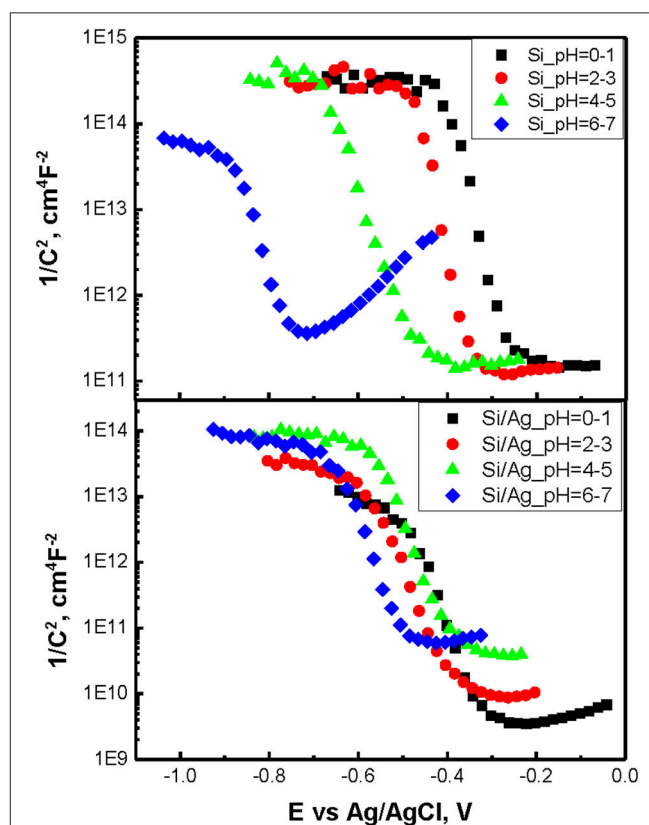


FIGURE 4 | Mott-Schottky measurements at 1,000 Hz.

of Si electrode. Resistivity of SiO_x layer predictably increases with shifting to positive potentials vs. Ag/AgCl reference due to growing layer thickness. As soon as first semicircle appear at higher frequencies (typically $>1,000$ Hz) it can only correspond to the processes at Si/SiO_x interface. This parallel RC element can be ascribed to the accumulation layer in Si resulting in downward bending of the valence and the conduction bands. Decreasing the radius of this semicircle with shifting to positive potentials is than well-explained by band flattening in p-doped silicon.

Notably both the SiO_x layer thickness and electrolyte potential are strongly affected by pH. With increasing pH of electrolyte OCP of the cell decreases reducing silicon oxidation rate (Figure 3B, Table 1). However, the radius of the second RC elements grows due to lower dissolution rate of silicon dioxide resulting in higher capacitance of the layer (Figure 3B).

Addition of silver particles to the system introduces a number of changes to the impedance spectra. First, Z'' at high frequencies strongly decreases implying lowering of the capacitance at Si/SiO_x interface. Secondly, the radius of the second semicircle greatly increases indicating larger thickness of SiO_x layer. These effects are associated with inhomogeneous nature of Ag/Si electrodes where both silver coated and uncoated regions contribute the impedance spectra. Probably Si/SiO_x/Ag/Ag₂O/H₂O₂ electrochemical chain provides smaller barrier as compared to direct electric double layer contact Si/SiO_x/H₂O₂. However, in case of low frequencies the depletion of charge carriers from Ag/Si results in limitation of carrier transport and SiO_x layer capacitance growth.

To determine flat band potentials of p-doped Si and Ag/Si electrodes Mott-Schottky measurements were performed at 1,000 Hz. The choice of the frequency was dictated by the necessity to attain depletion of the charge carriers while avoiding diffusion limitations. Resulting plots for different pH of etching solutions and derived flat band potential values are summarized

in Figure 4 and Table 1. One can see, the flat band potentials being pH dependent in case of etching of p-doped silicon converge into closely the same value in case of Ag/Si. On the other hand, OCP values stay very close in both p-Si and Ag/Si, with only small shift of OCP in case of Ag/Si. This effect corresponds well to smaller band bending and smaller capacitance of the interface layers. Thus, silver assisted chemical etching of silicon can be ascribed to facilitated transport through Si/SiO_x/Ag interface.

Total reflectance spectra of SiNW layers are presented in Figure 5. All samples exhibit a strong decrease of the total reflectance to 5–10% at the wavelength <850 nm in comparison to c-Si substrate (50%). At pH >3 , the total reflection spectra of nanowires have a very similar form with c-Si, since for a weak submicron length SiNW, the c-Si substrate has a significant effect on the reflection value. Also in this case, reflection peaks appear at 280 and 370 nm, which are associated with the c-Si direct band gap. Low total reflection of SiNW layers can be explained by the

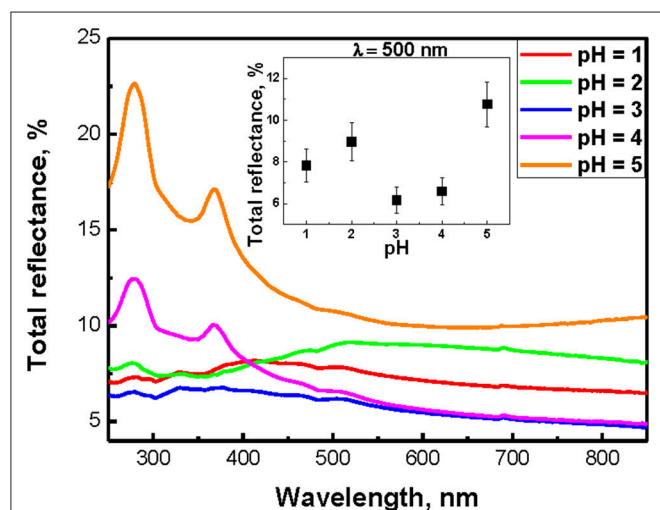


FIGURE 5 | Total reflectance spectra of SiNWs with different pH of $\text{H}_2\text{O}_2:\text{NH}_4\text{F}$; inset shows the dependence of total reflection of SiNWs from the pH value of $\text{H}_2\text{O}_2:\text{NH}_4\text{F}$.

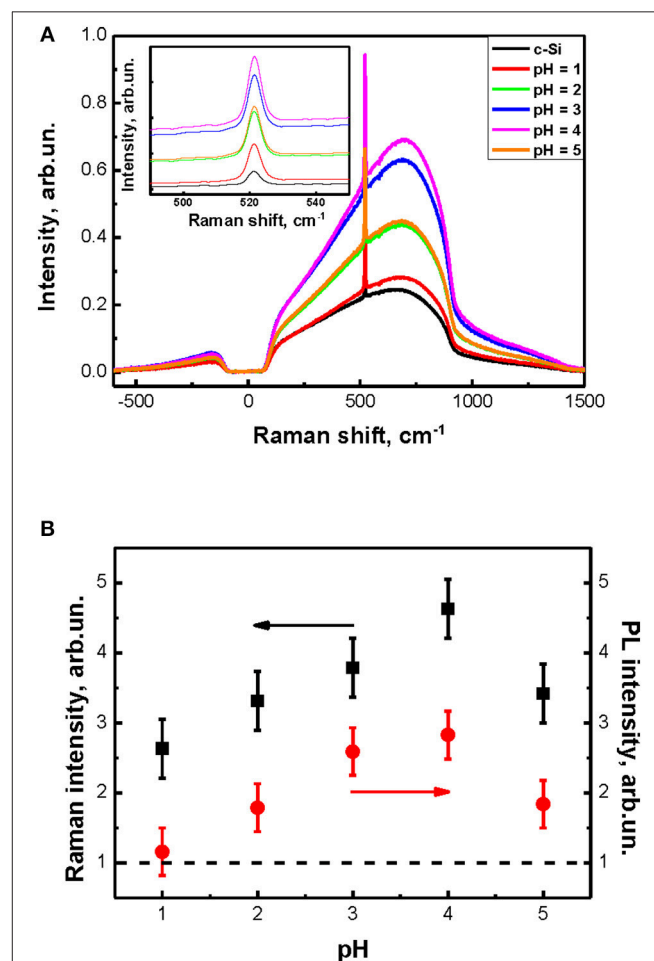


FIGURE 6 | (A) Spectra of interband PL and Raman scattering of c-Si substrate and SiNWs with different pH of $\text{H}_2\text{O}_2:\text{NH}_4\text{F}$; inset shows Raman scattering peaks of c-Si substrate and SiNWs with different pH of $\text{H}_2\text{O}_2:\text{NH}_4\text{F}$. **(B)** The dependence of intensities of Raman scattering and interband PL of SiNWs from the pH value of $\text{H}_2\text{O}_2:\text{NH}_4\text{F}$.

strong scattering and absorption of light in the visible region of the spectrum, which can lead to a partial localization of light in nanowires (Gonchar et al., 2012). The inset in **Figure 5** shows the dependence of the total reflection of SiNWs at 500 nm from the pH value of $\text{H}_2\text{O}_2\text{:NH}_4\text{F}$. It is seen that for this wavelength, all samples have the same low values of the total reflectance (5–10%).

The spectra of interband PL (broad peak) and Raman scattering (sharp peak at 520 cm^{-1}) of the c-Si substrate and a number of SiNW grown at different pH values are shown in **Figure 6A**. The inset in **Figure 6A** shows a close view of the Raman scattering peaks. SiNW's diameter is about 50–200 nm and far from the quantum confinement regime. That's why peaks and shapes of the interband PL and Raman scattering for all samples are similar to c-Si. At the same time the intensities of interband PL and Raman scattering for SiNWs increase strongly as opposed to corresponding value for c-Si. This effect can be explained by the light localization in such inhomogeneous optical medium as SiNW layers (Gonchar et al., 2014).

Figure 6B shows the calculated from **Figure 6A** dependence of SiNW's Raman scattering and interband PL intensities from the pH value. The signal intensity of the samples here was normalized to the signal intensity of c-Si substrate (dash line). Thus, the intensity of Raman scattering and interband PL increases by 3–5 times and 3 times, respectively, for all SiNWs layers in comparison with c-Si. Let's remember, that the shape and length of SiNWs is changed with the increasing of pH value of $\text{H}_2\text{O}_2\text{:NH}_4\text{F}$: the length is decrease and the shape is changing from vertical cylinders to pyramidal like structures (see **Figure 1**). Based on this, we can conclude that the intensity of Raman scattering and interband PL depends not only on the length of SiNW, but also on their shape.

CONCLUSION

The structural and optical properties of SiNWs, prepared by the metal assisted chemical etching method, where the commonly used hydrofluoric acid (HF) has been successfully replaced with ammonium fluoride (NH_4F), and their dependence from the pH of the etching $\text{H}_2\text{O}_2\text{:NH}_4\text{F}$ solutions were studied in detail for

the first time. It is shown that as the pH of $\text{H}_2\text{O}_2\text{:NH}_4\text{F}$ decrease, the shape of the nanowires changes from pyramidal to vertical. The length of SiNW arrays demonstrated non-linearly pH dependence. By impedance and Mott-Schottky measurements it was shown that the SiO_x layer thickness and electrolyte potential are strongly affected by pH. With increasing pH of electrolyte OCP of the cell decreases reducing silicon oxidation rate. Silver assisted chemical etching of silicon can be ascribed to facilitated charge carriers transport through Si/ SiO_x /Ag interface. All samples exhibit a strong decrease of the total reflectance to 5–10% at the wavelength $<800\text{ nm}$ in comparison to c-Si substrate. Also the intensities of interband PL and Raman scattering for SiNWs increase strongly as opposed to corresponding value for c-Si, but depends both from the length and the shape of SiNWs: they were larger for long pyramidal nanowires. This effect can be explained by the light localization in such inhomogeneous optical medium as SiNW layers. Thus, SiNW, manufactured using weakly toxic NH_4F , have great potential for applications in the field of photovoltaics, photonics, and sensorics.

AUTHOR CONTRIBUTIONS

KG and VK performed SiNWs fabrication, optical measurements, and data analysis. GZ performed the SEM measurements. AE performed impedance and Mott-Schottky measurements. KG and LO performed the general data analysis and discussion of the obtained data. All authors read and approved the final manuscript.

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REFERENCES

- Bai, F., Li, M., Huang, R., Song, D., Jiang, B., and Li, Y. (2012). Template-free fabrication of silicon micropillar/nanowire composite structure by one-step etching. *Nanosci. Res. Lett.* 7:557. doi: 10.1186/1556-276x-7-557
- Bertolini, J. C. (1992). Hydrofluoric acid: a review of toxicity. *J. Emergency Med.* 10, 163–168. doi: 10.1016/0736-4679(92)90211-B
- Brönstrup, G., Jahr, N., Leiterer, C., Csáki, A., Fritzsche, W., and Christiansen, S. (2010). Optical properties of individual silicon nanowires for photonic devices. *ACS Nano* 4, 7113–7122. doi: 10.1021/nn101076t
- Bruggeman, D. A. G. (1935). Berechnung verschiedener physikalischer konstanten von heterogenen substanzen. *Ann. Phys.* 24, 636–679. doi: 10.1002/andp.19354160705
- Cao, A., Sudhölter, E. J., and de Smet, L. C. (2015). Silicon nanowire-based devices for gas-phase sensing. *Sensors* 14, 245–271. doi: 10.3390/s140100245
- Chattopadhyay, S., Li, X., and Bohn, P. W. (2002). In-plane control of morphology and tunable photoluminescence in porous silicon produced by metal-assisted electroless chemical etching. *J. Appl. Phys.* 91, 6134–6140. doi: 10.1063/1.1465123
- Cui, Y., Wei, Q., Park, H., and Lieber, C. M. (2001). Nanowire nanosensors for highly sensitive and selective detection of biological and chemical species. *Science* 293, 1289–1292. doi: 10.1126/science.1062711
- Dawood, M. K., Tripathy, S., Dolmanan, S. B., Ng, T. H., Tan, H., and Lam, J. (2012). Influence of catalytic gold and silver metal nanoparticles on structural, optical, and vibrational properties of silicon nanowires synthesized by metal-assisted chemical etching. *J. Appl. Phys.* 112:073509. doi: 10.1063/1.4757009
- Dittrich, T., Rauscher, S., Timoshenko, V. Y., Rappich, J., Sieber, I., Flietner, H., et al. (1995). Ultrathin luminescent nanoporous silicon on n-Si: pH dependent preparation in aqueous NH_4F solutions. *Appl. Phys. Lett.* 67, 1134–1136. doi: 10.1063/1.114985
- Föll, H., Hartz, H., Ossei-Wusu, E., Carstensen, J., and Riemenschneider, O. (2010). Si nanowire arrays as anodes in Li ion batteries. *Phys. Status Solidi RRL* 4, 4–6. doi: 10.1002/pssr.200903344

- Georgobiani, V. A., Gonchar, K. A., Zvereva, E. A., and Osminkina, L. A. (2018). Porous silicon nanowire arrays for reversible optical gas sensing. *Phys. Stat. Sol. A* 215:1700565. doi: 10.1002/pssa.201700565
- Gonchar, K. A., Osminkina, L. A., Galkin, R. A., Gongalsky, M. B., Marshov, V. S., Timoshenko, V. Y., et al. (2012). Growth, structure and optical properties of silicon nanowires formed by metal-assisted chemical etching. *J. Nanoelectr. Optoelectr.* 7, 602–606. doi: 10.1166/jno.2012.1401
- Gonchar, K. A., Osminkina, L. A., Sivakov, V., Lysenko, V., and Timoshenko, V. Y. (2014). Optical properties of nanowire structures produced by the metal-assisted chemical etching of lightly doped silicon crystal wafers. *Semiconductors* 48, 1613–1618. doi: 10.1134/S1063782614120082
- Gonchar, K. A., Zubairova, A. A., Schleusener, A., Osminkina, L. A., and Sivakov, V. (2016). Optical properties of silicon nanowires fabricated by environment-friendly chemistry. *Nanosci. Res. Lett.* 11:357. doi: 10.1186/s11671-016-1568-5
- Gorostiza, P., Diaz, R., Kulandainathan, M. A., Sanz, F., and Morante, J. R. (1999). Simultaneous platinum deposition and formation of a photoluminescent porous silicon layer. *J. Electroanal. Chem.* 469, 48–52. doi: 10.1016/S0022-0728(99)00189-8
- Jiang, Y., Shen, H., Zheng, C., Pu, T., Wu, J., Rui, C., et al. (2017). Nanostructured multi-crystalline silicon solar cell with isotropic etching by HF/KMnO₄. *Phys. Status Solidi A* 214:1600703. doi: 10.1002/pssa.201600703
- Judge, J. S. (1971). A study of the dissolution of SiO₂ in acidic fluoride solutions. *J. Electrochem. Soc.* 118, 1772–1775. doi: 10.1149/1.2407835
- Kelzenberg, M. D., Turner-Evans, D. B., Kayes, B. M., Filler, M. A., Putnam, M. C., Lewis, N. S., et al. (2008). Photovoltaic measurements in single-nanowire silicon solar cells. *Nano Lett.* 8, 710–714. doi: 10.1021/nl072622p
- Li, X., and Bohn, P. W. (2000). Metal-assisted chemical etching in HF/H₂O₂ produces porous silicon. *Appl. Phys. Lett.* 77, 2572–2574. doi: 10.1063/1.1319191
- Morinaga, H., Futatsuki, T., Ohmi, T., Fuchita, E., Oda, M., and Hayashi, C. (1995). Behavior of ultrafine metallic particles on silicon wafer. *Surface J. Electrochem. Soc.* 142, 966–970. doi: 10.1149/1.2048569
- Nahidi, M., and Kolasinski, K. W. (2006). Effects of stain etchant composition on the photoluminescence and morphology of porous silicon. *J. Electrochem. Soc.* 153, C19–C26. doi: 10.1149/1.2129558
- Peng, K. Q., Hu, J. J., Yan, Y. J., Wu, Y., Fang, H., Xu, Y., et al. (2006). Fabrication of single-crystalline silicon nanowires by scratching a silicon surface with catalytic metal particles. *Adv. Funct. Mater.* 16, 387–394. doi: 10.1002/adfm.200500392
- Peng, K. Q., Lu, A. J., Zhang, R. Q., and Lee, S. T. (2008). Motility of metal nanoparticles in silicon and induced anisotropic silicon etching. *Adv. Funct. Mater.* 18, 3026–3035. doi: 10.1002/adfm.200800371
- Peng, K. Q., Yan, Y. J., Gao, S. P., and Zhu, J. (2002). Synthesis of large-area silicon nanowire arrays via self-assembling nanoelectrochemistry. *Adv. Mater.* 14, 1164–1167. doi: 10.1002/1521-4095(20020816)14:16<1164::AID-ADMA1164>3.0.CO;2-E
- Sivakov, V., Andrä, G., Gawlik, A., Berger, A., Plentz, J., Falk, F., et al. (2009). Silicon nanowire-based solar cells on glass: synthesis, optical properties, and cell parameters. *Nano Lett.* 9, 1549–1554. doi: 10.1021/nl803641f
- Sivakov, V. A., Bronstrup, G., Pecz, B., Berger, A., Radnoczi, G. Z., Krause, M., et al. (2010). Realization of vertical and zigzag single crystalline silicon nanowire architectures. *J. Phys. Chem. C* 114, 3798–3803. doi: 10.1021/jp909946x
- Stelzner, T., Pietsch, M., Andrä, G., Falk, F., Ose, E., and Christiansen, S. H. (2008). *Nanotechnology* 19:295203. doi: 10.1088/0957-4484/19/29/295203
- Wagner, R. S., and Ellis, W. C. (1964). Vapor-liquid-solid mechanism of single crystal growth. *Appl. Phys. Lett.* 4, 89–90. doi: 10.1063/1.1753975
- Wang, X., and Ozkan, C. S. (2008). Multisegment nanowire sensors for the detection of DNA molecules. *Nano Lett.* 8, 398–404. doi: 10.1021/nl071180e
- Yang, P., Yan, R., and Fardy, M. (2010). Semiconductor nanowire: what's next? *Nano Lett.* 10, 1529–1536. doi: 10.1021/nl100665r
- Zhang, M. L., Peng, K. Q., Fan, X., Jie, J. S., Zhang, R. Q., Lee, S. T., et al. (2008). Preparation of large-area uniform silicon nanowires arrays through metal-assisted chemical etching. *J. Phys. Chem. C* 112, 4444–4450. doi: 10.1021/jp077053o

Conflict of Interest Statement: The authors declare that the research was conducted in the absence of any commercial or financial relationships that could be construed as a potential conflict of interest.

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